

## Preparation and Optical Properties of Fullerene/Ferrocene Hybrid Hexagonal Nanosheets and Large-Scale Production of Fullerene Hexagonal Nanosheets

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**Abstract:** The supramolecular nanoarchitectures, C<sub>60</sub>/ferrocene nanosheets, were prepared by a simple liquid–liquid interfacial precipitation method and fully characterized by means of SEM, STEM, HRTEM, XRD, Raman and UV–vis–NIR spectra. The highly crystallized C<sub>60</sub>/ferrocene hexagonal nanosheets had a size of ca. 9 μm and the formulation C<sub>60</sub>(ferrocene)<sub>2</sub>. A strong charge-transfer (CT) band between ferrocene and C<sub>60</sub> was observed at 782 nm, indicating the presence of donor–acceptor interaction in the nanosheets. Upon heating the nanosheets to 150 °C, the CT band disappeared due to the sublimation of ferrocene from the C<sub>60</sub>/ferrocene hybrid, and C<sub>60</sub> nanosheets with an fcc crystal structure and the same shape and size as the C<sub>60</sub>/ferrocene nanosheets were obtained.

### Introduction

Because of their unique physical and chemical properties, fullerenes have tremendous potential as building blocks for new materials.<sup>1</sup> Recently, fullerene-based supramolecular nanoarchitectures such as nanowhiskers,<sup>2</sup> nanotubes,<sup>3</sup> nanorods,<sup>4</sup> nanowires,<sup>5</sup> and nanosheets<sup>6</sup> have attracted special interest

because of their unique chemical and physical properties and their possible application in the fields of materials and medical sciences. The unique properties are a result of the high symmetry of the nanoarchitectures and the presence of novel π-conjugated systems in them. Very recently, flower-like supramolecular assemblies were also prepared by using C<sub>60</sub> derivatives with long aliphatic chains.<sup>7</sup>

Fullerenes form a wide variety of donor–acceptor complexes with different classes of organic and organometallic donors.<sup>8</sup> These complexes show a wide range of physical properties, including metallic,<sup>9</sup> photoconducting,<sup>10</sup> and unusual magnetic properties.<sup>11</sup> Ferrocene (Fc) is one such very important donor because of its strong electron-donating ability and high stability under redox conditions. Therefore, supramolecular hybrids formed by the combination of fullerene and Fc are expected to have fascinating properties. Indeed, C<sub>60</sub>/Fc hybrid molecules in solution<sup>12–14</sup> and liquid crystals with supramolecular organized structures<sup>15–17</sup> have been reported to show interesting properties.

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- (1) Hirsch, A.; Brettreich, M. *Fullerenes*; Wiley-VCH Verlag GmbH: KGaA: Weinheim, Germany, 2005.
- (2) (a) Miyazawa, K.; Kuwasaki, Y.; Obayashi, A.; Kuwabara, M. *J. Mater. Res.* **2002**, *17*, 83–88. (b) Miyazawa, K.; Hamamoto, K.; Nagata, S.; Suga, T. *J. Mater. Res.* **2003**, *18*, 1096–1103.
- (3) (a) Liu, H.; Li, Y.; Jiang, L.; Luo, H.; Xiao, S.; Fang, H.; Li, H.; Zhu, D.; Yu, D.; Xu, J.; Xiang, B. *J. Am. Chem. Soc.* **2002**, *124*, 13370–13371. (b) Minato, J.; Miyazawa, K. *Diamond Relat. Mater.* **2006**, *15*, 1143–1146. (c) Ringer, C.; Miyazawa, K.; Awane, T. *Trans. Mat. Res. Jpn.* **2007**, *32*, 1011–1014.
- (4) (a) Wang, L.; Liu, B.; Liu, D.; Yao, M.; Hou, Y.; Yu, S.; Cui, T.; Li, D.; Zou, G.; Iwasiewicz, A.; Sundqvist, B. *Adv. Mater.* **2006**, *18*, 1883–1888. (b) Jin, Y.; Curry, R. J.; Sloan, J.; Hatton, R. A.; Chong, L. C.; Blanchard, N.; Solojan, V.; Kroto, H. W.; Silva, S. R. P. *J. Mater. Chem.* **2006**, *16*, 3715–3720. (c) Tsuchiya, T.; et al. *J. Am. Chem. Soc.* **2008**, *130*, 450–451.
- (5) (a) Geng, J.; Zhou, W.; Skelton, P.; Yue, W.; Kinloch, I. A.; Wimdle, A. H.; Johnson, B. F. G. *J. Am. Chem. Soc.* **2008**, *130*, 2527–2534. (b) Malik, S.; Fujita, N.; Muhopadhyay, P.; Goto, Y.; Kaneko, K.; Ikeda, T.; Shinkai, S. *J. Mater. Chem.* **2007**, *17*, 2454–2458.
- (6) (a) Sathish, M.; Miyazawa, K. *J. Am. Chem. Soc.* **2007**, *129*, 13816–13817. (b) Wang, L.; et al. *Appl. Phys. Lett.* **2007**, *91*, 103112–103113.

- (7) Nakanishi, T.; Ariga, K.; Michinobu, T.; Yoshida, K.; Takahashi, H.; Teranishi, T.; Mohwald, H.; Kurth, D. G. *Small* **2007**, *3*, 2019–2023.
- (8) Konarev, D. V.; Lyubovskaya, R. N.; Drichko, N. V.; Yudanva, E. I.; Shul'ga, Y. M.; Litvinov, A. L.; Semkin, V. N.; Tarasov, B. P. *J. Mater. Chem.* **2000**, *10*, 803–918.
- (9) Moriyama, H.; Kobayashi, H.; Kobayashi, A.; Watanabe, T. *Chem. Phys. Lett.* **1995**, *238*, 116–121.
- (10) Konarev, D. V.; Kovalevsky, A. Y.; Khasanov, S. S.; Saito, G.; Lopatin, D. V.; Umrikhin, A. V.; Otsuka, A.; Lyubovskaya, R. N. *Eur. J. Inorg. Chem.* **2006**, *188*, 1–1895.
- (11) (a) Stephens, P. W.; Cox, D.; Lauher, J. W.; Mihaly, L.; Wiley, J. B.; Allemand, P.-M.; Hirsch, A.; Holczer, K.; Li, Q.; Thompson, J. D.; Wudl, F. *Nature* **1992**, *355*, 331–332. (b) Wang, H.; Zhu, D. *Solid State Commun.* **1995**, *95*, 295–299.

As a crystalline state, C<sub>60</sub>/Fc hybrid was first reported by Crane et al. in 1992;<sup>18</sup> they determined the crystal structure by assuming motionless C<sub>60</sub> molecules. These authors inferred that Fc does not donate fully its electron to C<sub>60</sub> and that contribution of the weak intermolecular charge-transfer (CT) interaction to the stability of the hybrid is very small.<sup>18</sup> However, the electronic properties of the hybrid are not still clear.

We herein report a simple method for the preparation of C<sub>60</sub>/Fc hybrid hexagonal nanosheets with a size of ca. 9 μm by a liquid–liquid interfacial precipitation (LLIP) method. This method was introduced by our group for the preparation of C<sub>60</sub> nanowhiskers at a toluene/isopropyl alcohol (IPA) interface.<sup>2a</sup> In this paper, we also discuss the CT interaction between C<sub>60</sub> and Fc in nanosheets on the basis of CT-band observations in the solid state. Interestingly, the CT band disappears after heat treatment, which is accompanied by the sublimation of Fc. After the heat treatment, face-centered cubic (fcc) C<sub>60</sub> nanosheets are observed to be left behind.

## Experimental Section

The C<sub>60</sub>/Fc nanosheets were prepared by the LLIP method, in which an interface containing C<sub>60</sub> and excess Fc was formed between IPA and toluene. An amount of 3 mL of C<sub>60</sub>-saturated toluene was taken in a 10 mL glass bottle and 120 mg of Fc was added to this solution. After 10-min ultrasonication at room temperature, the C<sub>60</sub>/Fc solution was cooled to 10 °C in an ice–water bath. To this toluene solution, 3 mL of IPA was added slowly, and the mixture was maintained at 10 °C for 5 min without disturbing the interface. The resulting mixture was stored at 10 °C for 24 h for the growth of C<sub>60</sub>/Fc nanosheets. C<sub>60</sub>/Fc nanosheets were obtained by filtering the solution and were heated at 80 °C for 15 min to remove the excess Fc.<sup>18</sup>

The structure and morphology of the obtained nanosheets were characterized using scanning transmission electron microscopy (STEM; JEOL JEM-2100F, 200 kV), field emission scanning electron microscopy (FE-SEM; Hitachi S-4800, 15 kV, FE-SEM; JEOL JSM-6700F, 5 kV), and a micro-Raman system (NRS-3100, JASCO, Japan) equipped with a semiconducting laser with a wavelength of 532 nm. Diffuse reflectance spectra of the nanosheets were measured by a UV–vis–NIR spectroscope (V-570, JASCO, Japan) equipped with an integrating sphere. The prepared C<sub>60</sub>/Fc nanosheets were taken on a quartz plate and dried at room temperature prior to the measurement of the spectra. UV–vis–NIR spectra of C<sub>60</sub> and Fc powder samples were also measured in a similar manner.

Calculations were carried out for ascertaining the structural stability and determining the electronic states within the local density approximation (LDA)<sup>19</sup> to the density functional theory (DFT). A plane wave basis set and a pseudopotential scheme were adopted for the crystal system. For determining the CT excitation energy, we used the Slater transition (ST) method and the modified

linear response (MLR) scheme for the time-dependent (TD) DFT.<sup>20</sup> All the calculations were performed using the ABINIT code.<sup>21</sup> The calculation details are described in Supporting Information.

## Results and Discussion

**Preparation and Structure of C<sub>60</sub>/Fc Nanosheets.** In the preparation of C<sub>60</sub>/Fc nanosheets by the LLIP method through the formation of an interface between IPA and toluene solution containing C<sub>60</sub> and Fc, it is very interesting to note that the concentration of Fc in toluene must be more than 30 mg/mL. When the concentration of Fc is less than 20 mg/mL, C<sub>60</sub> nanowhiskers are formed and Fc molecules just cover the external surface of the C<sub>60</sub> nanowhiskers, as explained in our previous report.<sup>22</sup> This indicates that, at low Fc concentrations, Fc molecules cannot enter the C<sub>60</sub> crystals due to the strong interaction between the C<sub>60</sub> molecules. Consequently, C<sub>60</sub> nanowhiskers are surrounded by Fc molecules. However, at high Fc concentrations, Fc and C<sub>60</sub> form precipitates with quite different morphologies. These precipitates are found to be C<sub>60</sub>/Fc hybrid hexagonal nanosheets in the present study.

SEM images of C<sub>60</sub>/Fc precipitates prepared at high Fc concentrations are shown in Figure 1. These representative images show the hexagonal morphology of the precipitates, which are highly transparent to electron beams. The average size of the precipitates is 9.1 ± 6.2 μm and the thickness is about 250–550 nm, confirming the formation of hexagonal nanosheets. It is very interesting to note that many self-standing nanosheets are observed in the SEM images in Figure 1, indicating rigid crystalline materials. The morphology of these nanosheets is very similar to that of hexagonal C<sub>60</sub> nanosheets prepared solely from C<sub>60</sub> molecules using the LLIP method, which was reported by our group recently.<sup>6</sup>

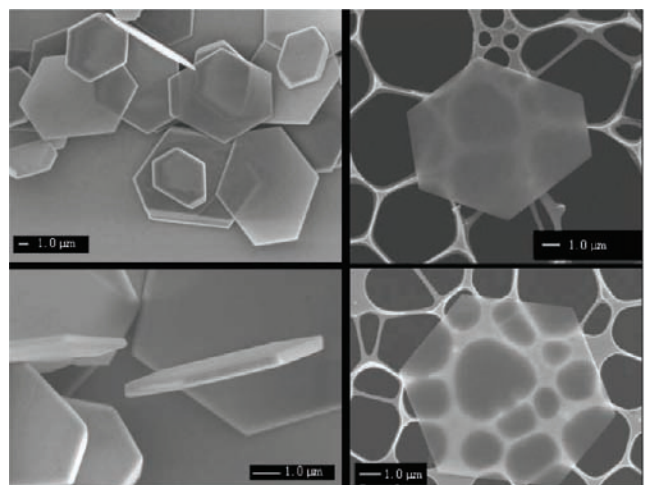
To locate the doped Fc in the nanosheets, a STEM mapping analysis was carried out (Figure 2). Carbon and iron atoms were detected in the nanosheets, as shown in Figure 2, panels b and c, respectively. The distribution and concentration of carbon and iron were uniform in the nanosheets. This result confirms the fine dispersion of iron atoms in the nanosheets, indicating that the nanosheets are C<sub>60</sub>/Fc nanosheets.

Figure 3 shows an HR-TEM image of C<sub>60</sub>/Fc nanosheets. A clear lattice structure is observed. The distance between lattices, *d*, is calculated to be 0.90 nm.

X-ray diffraction (XRD) patterns of the C<sub>60</sub> powder and C<sub>60</sub>/Fc nanosheets are shown in Figure 4. The XRD pattern of the C<sub>60</sub>/Fc nanosheets is quite different from that of the C<sub>60</sub> powder, but identical to that of triclinic C<sub>60</sub>(Fc)<sub>2</sub> prepared directly from a solution of C<sub>60</sub> in molten Fc.<sup>23</sup> This observation indicates that the C<sub>60</sub>/Fc nanosheets contain C<sub>60</sub>(Fc)<sub>2</sub> units, which have two Fc molecules near every C<sub>60</sub> molecule.<sup>18</sup> The inset in Figure 4 shows the selected-area electron diffraction pattern of a C<sub>60</sub>/Fc nanosheet. The observed pattern is indexed for the (010) and (2̄10) planes. The *d* value calculated from the HR-TEM image in Figure 3 is indicated as the (010) plane.

- (12) Hauke, F.; Hirsch, A.; Liu, S. G.; Echegoyen, L.; Swartz, A.; Luo, C. P.; Guldi, D. M. *ChemPhysChem* **2002**, *3*, 195–205.
- (13) Guldi, D. M.; Rahman, G. M. A.; Marczak, R.; Matsuo, Y.; Yamana, M.; Nakamura, E. *J. Am. Chem. Soc.* **2006**, *128*, 9420–9427.
- (14) Marczak, R.; Wielopolski, M.; Gayathri, S. S.; Guldi, D. M.; Matsuo, Y.; Matsuo, K.; Tahara, K.; Nakamura, E. *J. Am. Chem. Soc.* **2008**, *130*, 16207–16215.
- (15) Even, M.; Heinrich, B.; Guillon, D.; Guldi, D. M.; Prato, M.; Deschenaux, R. *Chem.–Eur. J.* **2001**, *7*, 2595–2604.
- (16) Campidelli, S.; Vazquez, E.; Milic, D.; Prato, M.; Barbera, J.; Guldi, D. M.; Marcaccio, M.; Paolucci, D.; Paolucci, F.; Deschenaux, R. *J. Mater. Chem.* **2004**, *14*, 1266–1272.
- (17) Matsuo, Y.; Muramatsu, A.; Kamikawa, Y.; Kato, T.; Nakamura, E. *J. Am. Chem. Soc.* **2006**, *128*, 9586–9587.
- (18) Crane, J. D.; Hitchcock, P. B.; Kroto, H. W.; Taylor, R.; Walton, D. R. M. *J. Chem. Soc., Chem. Commun.* **1992**, 1764–1765.
- (19) Perdew, J. P.; Wang, Y. *Phys. Rev. B* **1992**, *45*, 13244–13249.

- (20) (a) Hu, C.; Sugino, O.; Miyamoto, Y. *Phys. Rev. A* **2006**, *74*, 032508.  
(b) Hu, C.; Sugino, O. *J. Chem. Phys.* **2007**, *126*, 074112.
- (21) Gonze, X.; Beuken, J.-M.; Caracas, R.; Detraux, F.; Fuchs, M.; Rignanese, G.-M.; Sindic, L.; Verstraete, M.; Zerah, G.; Jollet, F.; Torrent, M.; Roy, A.; Mikami, M.; Ghosez, Ph.; Raty, J.-Y.; Allan, D. C. *Comput. Mater. Sci.* **2002**, *25*, 478–492.
- (22) Wakahara, T.; Sathish, M.; Miyazawa, K.; Sasaki, T. *Nano* **2008**, *3*, 351–354.
- (23) Espeau, P.; Barrio, M.; Lopez, D. O.; Tamarit, J. L.I.; Ceolin, R.; Allouchi, H.; Agafonov, V.; Masin, F.; Szwarc, H. *Chem. Mater.* **2002**, *14*, 321–326.



**Figure 1.** SEM images (left, on the grids; right, on the silicon wafer) and size distribution histograms of  $C_{60}/Fc$  nanosheets. (The size of the sheets is defined as the length of the longest diagonal line of the sheet.)

Recently, the electronic interaction between  $C_{60}$  and Fc molecules has been clarified; for example, it has been proposed that supramolecular organization in fullerene-Fc liquid crystals gives rise to a smectic A phase.<sup>15,16</sup>  $C_{60}$  imposes the arrangement of other molecular moieties to form a partial bilayer and the Fc moieties are localized between the  $C_{60}$  units and the dendric core.<sup>15,16</sup> In the case of  $C_{60}/Fc$  nanosheets, each  $C_{60}$  is surrounded by two Fc molecules, showing a triclinic  $C_{60}(Fc)_2$  structure.<sup>18</sup>

**Raman Spectra of  $C_{60}/Fc$  Nanosheets.** Figure 5 shows the Raman spectra of the prepared  $C_{60}/Fc$  nanosheets, pristine  $C_{60}$  powder, and Fc powder. The Raman spectrum of the prepared  $C_{60}/Fc$  nanosheets is almost identical to the summed spectra of the pristine  $C_{60}$  powder and Fc powder. The Raman lines observed in the case of the prepared  $C_{60}/Fc$  nanosheets are a combination of those of the  $C_{60}$  molecule (269, 430, 491, 709, 1422, 1465, and  $1568\text{ cm}^{-1}$ )<sup>24</sup> and the Raman lines of the Fc molecule (317 and  $1105\text{ cm}^{-1}$ ). These observations indicate that the nanosheets contain  $C_{60}$  and Fc.

**Optical Properties of  $C_{60}/Fc$  Nanosheets.** Figure 6 shows the diffuse reflectance spectra of the  $C_{60}/Fc$  nanosheets and  $C_{60}$  powder. The extra spectrum is obtained by the subtraction of a

normalized  $C_{60}$  spectrum from the spectrum of the  $C_{60}/Fc$  nanosheets, which can be attributed to the CT band of  $C_{60}/Fc$  nanosheets with absorption maximum at 782 nm, since CT interaction between  $C_{60}$  and Fc has been frequently reported in solution and liquid-crystalline.<sup>12–16</sup> Interestingly, the disappearance of the 782 nm band when the nanosheets are heated to 150 °C confirms that the band is a CT band. The CT transition energy corresponding to the maximum at 782 nm is estimated as 1.59 eV. It has been reported that the position of the CT band of  $C_{60}$  complexes depends on the ionization potentials of the donors;<sup>25</sup> from reported linear relation, the observed CT band of 1.59 eV for  $C_{60}/Fc$  nanosheets can be obtained using the ionization potential of 7.5 eV for Fc.

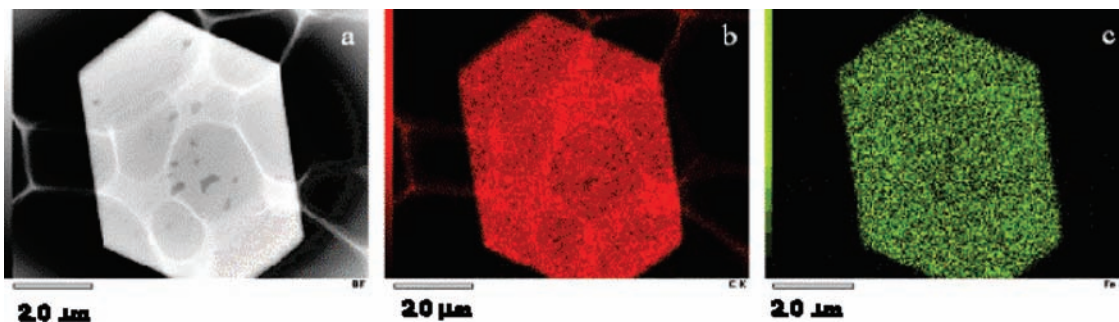
**Theoretical Calculations.** By considering the unit cell structure determined by XRD in this study, we perform the structural optimization of the  $C_{60}(Fc)_2$  crystal. The relaxed geometry is shown in Figure 7 and is consistent with the structure reported on the basis of XRD analysis of single crystal.<sup>18</sup> The nearest distance between a  $C_5H_5$  ring of the Fc and a pentagonal face of the  $C_{60}$  is calculated to be 3.23 Å, and the eclipsed Fc is displaced with respect to the pentagonal center by 1.23 Å. The highest occupied molecular orbital (HOMO) mainly consists of Fe 3d orbitals of the two Fc molecules, while the triply degenerated lowest unoccupied molecular orbital (LUMO) mainly comprises  $t_{1u}$  orbitals of  $C_{60}$  and a small part of Fe 3d orbitals of the two Fc molecules (see Supporting Information). Thus, the electronic transition from the HOMO to LUMO is the cause of the CT excitation. At the DFT Kohn–Sham level, intramolecular valence excitations in Fc and  $C_{60}$  are calculated to be 2.4 and 1.6 eV, respectively, which is consistent with experimental absorption edges around 500 and 700 nm. The ST method and (TD)-DFT-MLR scheme give the CT excitation energy values as 1.005 and 1.016 eV, respectively, which can be reasonably assigned to the broad absorption with peak around 800 nm and edge at 1000 nm in the diffuse reflectance spectra of the nanosheets.

**Conversion of  $C_{60}/Fc$  Nanosheets to  $C_{60}$  Nanosheets.** Subsequent to the preparation of  $C_{60}$  nanosheets, they have attracted great attention for their potential application to nanodevices and superhard materials.<sup>6</sup> However, synthetic methods continue to have limitations and the sizes and shapes of the nanosheets are not fully controllable. Recently, it has been reported that  $C_{60}(Fc)_2$  solids thermally decompose into fcc  $C_{60}$  crystals and Fc vapor at elevated temperature.<sup>23</sup> Therefore, the preparation of  $C_{60}$  nanosheets from hexagonal  $C_{60}/Fc$  nanosheets is an interesting challenge. Figure 8 shows SEM images of the thermal decomposition products of  $C_{60}/Fc$  nanosheets. These representative images show that the hexagonal morphology is retained even after the heat treatment; in addition, the size is also almost unchanged. The Raman spectrum of the heat-treated nanosheets shows the disappearance of the Fc lines, indicating the conversion to hexagonal  $C_{60}$  nanosheets.

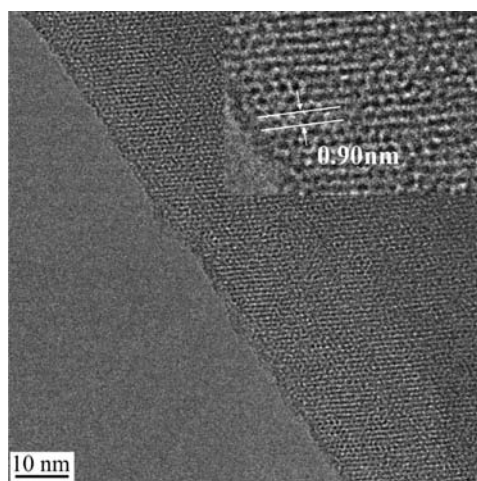
To investigate the crystal structure of the produced nanosheets, we carried out XRD measurements. The XRD patterns of the nanosheets, shown in Figure 4c, are quite different from those of  $C_{60}/Fc$  nanosheets, but almost identical to the patterns of fcc  $C_{60}$  powder, indicating that the nanosheets obtained are hexagonal  $C_{60}$  nanosheets. In Figure 4c, we observe an additional peak (indicated by \*); since this peak is not related to fcc  $C_{60}$ ,

(24) Kuzmany, H.; Pfeiffer, R.; Hulman, M.; Kramber, C. *Philos. Trans. R. Soc. London, Ser. A* **2004**, *362*, 2375–2406.

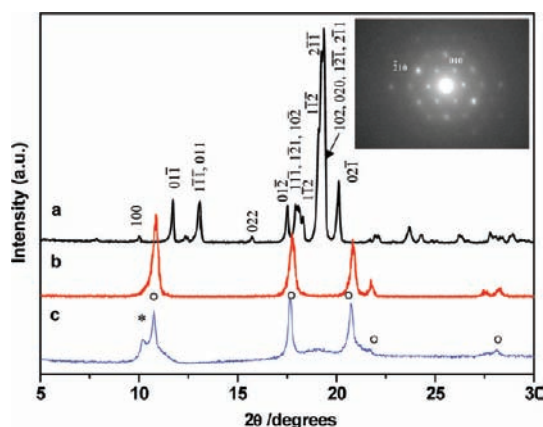
(25) Konarev, D. V.; Lyubovskaya, R. N.; Drichko, N. V.; Semkin, V. N.; Graja, A. *Chem. Phys. Lett.* **1999**, *314*, 570–576.



**Figure 2.** (a) STEM image of a  $C_{60}/Fc$  nanosheet and (b and c) STEM mapping images (b, carbon; c, iron) of  $C_{60}/Fc$  nanosheet.



**Figure 3.** HRTEM images of  $C_{60}/Fc$  nanosheets. Inset: magnified images.

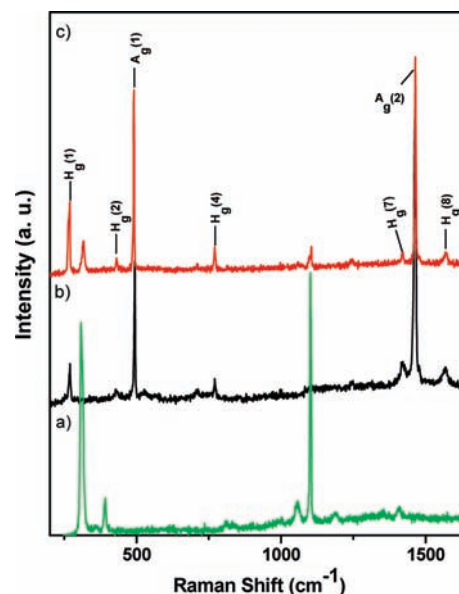


**Figure 4.** XRD patterns of (a)  $C_{60}/Fc$  nanosheets, (b)  $C_{60}$  powder, and (c) fcc  $C_{60}$  nanosheets. The inset is the selected-area electron diffraction pattern of  $C_{60}/Fc$  nanosheets. The peaks indicated by open circles correspond to the fcc structure of  $C_{60}$ .

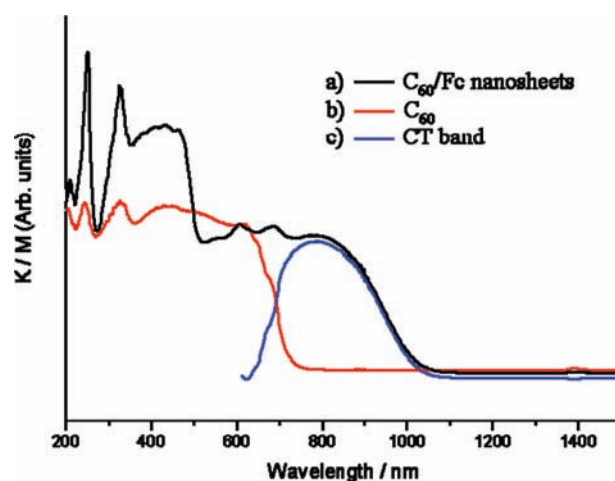
it may be associated with defects and stacking faults caused by the escape route of Fc molecules.<sup>26</sup>

These results indicate that triclinic  $C_{60}/Fc$  nanosheets are converted to fcc  $C_{60}$  nanosheets by the heat treatment, with the hexagonal morphology being retained through the conversion. This is in contrast to the formation of fibrous superstructures of  $C_{60}$  from the  $C_{60}/Fc$  system that has been reported by Shinkai and co-workers.<sup>5b</sup> There are two methods that have been

(26) Vaughan, G. B. M.; Chabre, Y.; Dubois, D. *Europhys. Lett.* **1995**, *31*, 525–530.

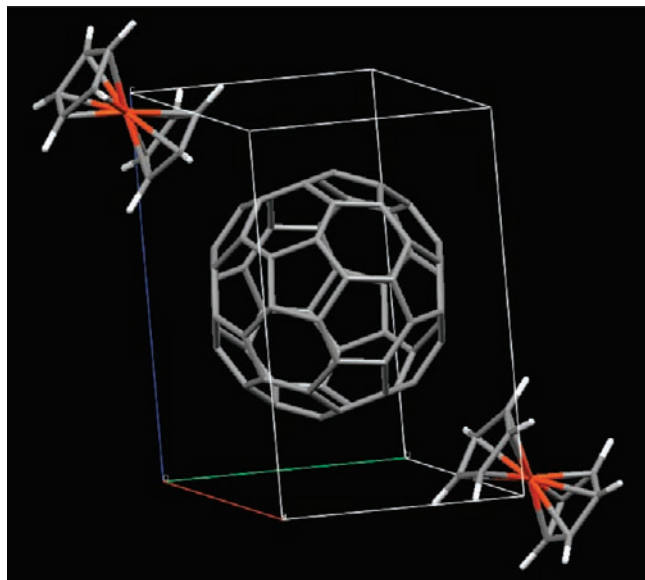


**Figure 5.** Raman spectra of (a) ferrocene, (b) pristine  $C_{60}$  powder, and (c) the  $C_{60}/Fc$  nanosheets.



**Figure 6.** Diffuse reflectance UV–vis–NIR spectra (K/M: Kubelka–Munk function) of (a) the  $C_{60}/Fc$  nanosheets, (b) pristine  $C_{60}$  powder, and (c) subtracted spectrum (a – b). Spectra a and b are normalized at 610 nm.

presented for the formation of  $C_{60}$  nanosheets, but it is very difficult to prepare  $C_{60}$  nanosheets on a large scale using these methods. In one method,  $C_{60}$  nanosheets are prepared by the precipitation method using  $CCl_4$  as a solvent; the solubility of  $C_{60}$  is low in this solvent.<sup>6a</sup> In the other method, the preparation is by the slow evaporation of *m*-xylene solution of  $C_{60}$  on

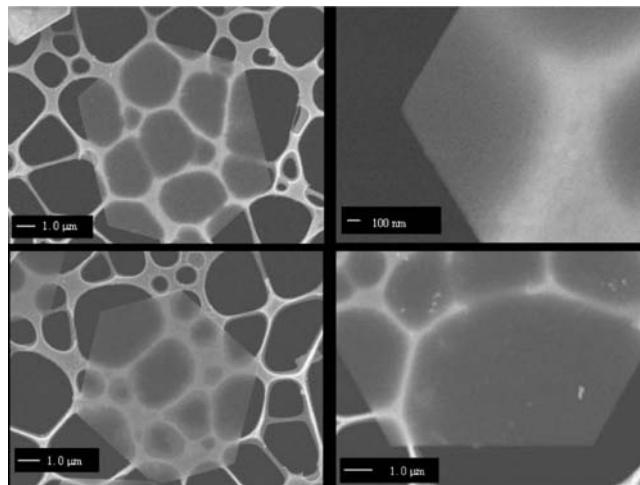


**Figure 7.** Relaxed geometry of  $C_{60}(\text{ferrocene})_2$  by ab initio DFT-LDA calculation.

substrates.<sup>6b</sup> The method for the preparation of hexagonal  $C_{60}$  nanosheets presented here is a very simple LLIP method using toluene/IPA and involving heat treatment at 150 °C. This method shows potential for the large-scale production of fcc  $C_{60}$  nanosheets with a range of sizes up to ca. 9  $\mu\text{m}$ .

### Conclusions

We have succeeded in the construction of supramolecular nanoarchitectures,  $C_{60}/\text{Fc}$  hexagonal nanosheets, with an average size ranging up to 9  $\mu\text{m}$  by a simple LLIP method. Diffuse reflectance spectra show a strong CT band at 782 nm, from which the CT transition energy is estimated as 1.59 eV. The presence of the CT band allows us to consider that the nearest  $C_{60}-\text{Fc}$  pair in the  $C_{60}/\text{Fc}$  hybrid interacts, and this interaction



**Figure 8.** SEM images of the  $C_{60}$  nanosheets.

is possibly the driving force for the formation of  $C_{60}/\text{Fc}$  nanosheets. These results are also supported by the theoretical calculations. The  $C_{60}/\text{Fc}$  hexagonal nanosheets can be converted to fcc  $C_{60}$  hexagonal nanosheets by heat treatment at 150 °C, at which both components  $C_{60}$  and Fc are thermally stable. The successful preparation of nanoarchitectures based on the CT interaction is expected to be an important stepping stone to the fabrication of novel fullerene nanodevices and to lead to a new field pertaining to fullerene engineering in nanoscience.

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**Supporting Information Available:** Complete refs<sup>4c</sup> and<sup>6b</sup> MO calculation details; optical microscope images; IR spectrum of nanosheets; HOMO–LUMO of  $C_{60}(\text{Fc})_2$ . This material is available free of charge via the Internet at <http://pubs.acs.org>.

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